



Ordered mesoporous CeO₂-TiO₂ composites: Highly efficient photocatalysts for the reduction of CO₂ with H₂O under simulated solar irradiation

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ABSTRACT

Ordered mesoporous CeO₂-TiO₂ composites with 2D hexagonal structure and varied compositions were synthesized through a nanocasting route using ordered mesoporous SBA-15 as the template. X-ray diffraction, nitrogen adsorption-desorption, transmission electron microscopy, X-ray photoelectron spectroscopy and UV-vis diffuse reflectance spectra analysis techniques were used to characterize the samples. It is observed that the obtained CeO₂-TiO₂ composites have ordered 2D hexagonal mesostructures with high specific surface area and hierarchical porosity. Introduction of CeO₂ species can effectively extend the spectral response from UV to visible area and enhance the surface chemisorbed oxygen of the ordered mesoporous TiO₂. Due to the peculiar composition and structural characteristics, these ordered mesoporous CeO₂-TiO₂ composites exhibited excellent photocatalytic activity in the reduction of CO₂ with H₂O under simulated solar irradiation.

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1. Introduction

The increasing atmospheric concentration of carbon dioxide (CO₂) due to fossil fuel combustion has raised serious concerns about global warming [1,2]. Solar photocatalytic conversion of CO₂ with H₂O into hydrocarbon fuels by semiconductors has sparked a new sustainable development path since it would help reduce atmospheric CO₂ levels and partly fulfill energy demands [3–5]. Among various oxide semiconductor photocatalysts, TiO₂ is one of the most common one due to its powerful oxidation properties, non-toxicity, low cost and long-term photostability [6–8]. However, the activity of pure TiO₂ in the photocatalytic reduction of CO₂ with H₂O is not high enough for practical use. To increase the photoactivity of TiO₂, several approaches have been suggested, such as reducing particle size [8,9], loading of metal [10–14], and dispersing on various supports [5,15–18]. These approaches have resulted in improvements of the catalytic performance of TiO₂-based photocatalysts but only to a limited extent. Recent progress shows that TiO₂ coupled with other semiconductors with suitable matching of band levels, like SnO₂ [19–22], WO₃ [23–25], CdS [26–28], Fe₂O₃ [29,30], CeO₂ [31,32], and ZrO₂ [33], can greatly increase the photocatalytic efficiency by increasing the charge separation and extending the energy range of photoexcitation. Among

them, CeO₂-TiO₂ composite is one of the most promising since the introduction of CeO₂ not only maintains their original structural features but also endows the resultant composite with additional properties [34,35].

In recent years, ordered mesostructures of metal oxides and their composites have attracted a great deal of attention due to their peculiar properties and wide potential applications [36]. Design of photocatalysts with well-defined mesoporous structure is a promising way to achieve high photocatalytic activity since the ordered mesopore channels facilitate fast intraparticle molecular transfer. While the large surface area may enhance the light harvesting and the adsorption for reactant molecules [37]. Although ordered mesoporous TiO₂ has already been widely investigated, only limited studies have been performed on the ordered mesoporous TiO₂ coupled semiconductor photocatalysts, especially for CeO₂-TiO₂ system. Recently, Yu et al. [38] reported the synthesis of ordered mesoporous CeO₂ doped TiO₂ photocatalyst (Ce/Ti molar ratio = 0.05) by an evaporation-induced self-assembly (EISA) method, their result showed the doping CeO₂ species into TiO₂ framework not only improve the thermal stability of the ordered mesoporous structure but also effectively extend the photo-response of TiO₂ to the visible-light region. However, to the best of our knowledge, there is no report on the synthesis of the ordered mesoporous CeO₂-TiO₂ composites with varied compositions and their activity toward photocatalytic CO₂ reduction.

In this work, we report a new approach to prepare ordered mesoporous CeO₂-TiO₂ composites with 2D hexagonal structure

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and varied compositions based on the nanocasting route. The synthesis was achieved using ordered mesoporous SBA-15 as a template, titanium tetrachloride and cerium nitrate were respectively used as the Ti source and Ce precursor. The structural properties of the obtained materials were analyzed by X-ray diffraction (XRD), nitrogen adsorption-desorption, transmission electron microscopy (TEM), X-ray photoelectron spectroscopy (XPS) and UV-vis diffuse reflectance spectra (UV-vis), their photocatalytic activity was examined toward the reduction of CO_2 with H_2O under simulated solar irradiation. It was found that the induced CeO_2 significantly enhanced the photoactivity of ordered mesoporous TiO_2 . To the best of our knowledge, this is the first time to report the enhanced photoactivity of ordered mesoporous CeO_2 - TiO_2 composites to the reduction of CO_2 with H_2O .

2. Experimental

2.1. Chemicals

Pluronic P123 ($\text{Mw} = 5800$, $\text{EO}_{20}\text{PO}_{70}\text{EO}_{20}$) was purchased from Aldrich. Tetraethoxysilane (TEOS), hydrochloric acid, glycerol, sodium hydroxide, acetone, cerium nitrate hexahydrate ($\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$), and titanium tetrachloride (TiCl_4) were purchased from Shanghai Chemical Corp. All chemicals were used as received without further purification.

2.2. Catalyst preparation

Ordered mesoporous silica SBA-15 template with rod-like morphology was synthesized as the reported procedure except enlarging the amount by ten times [39]. Ordered mesoporous CeO_2 - TiO_2 composites were synthesized using the hard template method with the help of acetone. In a typical preparation process, a total 12 mmol of TiCl_4 and $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ with different Ce/Ti molar ratios (0.5, 1.0, 2.0) in sequence was dissolved into 15 mL of acetone. After the solution became clear, 2 g of SBA-15 hard template was added and the mixture was stirred for 2 hours in a 100 mL beaker, then the mixture was transferred to a clean flat Petri dish (diameter: 8 cm) and the solvent was evaporated. In order to achieve higher loadings, the above dried hybrid powder was calcined at 200°C for 6 h to decompose the metal precursors, and then the impregnation step was repeated, but the amount of precursors was reduced to 2/3 compared to the first step. The resulting samples were calcined in air at 500°C for 4 h with a heating ramp of 1 K min^{-1} to completely decompose the inorganic precursors. Finally, the silica template was removed using a 2 M NaOH aqueous solution at room temperature and this etching process was repeated three times. The obtained CeO_2 - TiO_2 composites with different initial Ce/Ti molar

ratios (0.5, 1.0, 2.0) are denoted as Mes-CeTi-0.5, Mes-CeTi-1.0, and Mes-CeTi-2.0, respectively. For comparison, ordered mesoporous TiO_2 (Mes- TiO_2) and mesoporous CeO_2 (Mes- CeO_2) were also prepared by the same method.

2.3. Catalyst characterization

Low-angle X-ray diffraction patterns were collected in θ -2 θ mode using Rigaku D/MAX-2550VB/PC diffractometer ($\text{CuK}\alpha_1$ radiation, $\lambda = 1.5406\text{ \AA}$), operated at 40 kV and 200 mA (scanning step: 0.02° per second). Wide-angle XRD patterns were collected in the same mode, but operated at 100 mA. Transmission electron microscope (TEM) images were done using a JEOL JEM-2010 electron microscope with an acceleration voltage of 200 kV. Nitrogen sorption isotherms were measured at -196°C on a Micromeritics ASAP 2000 apparatus. The diffuse reflectance spectra of the samples over a range of 200–800 nm were recorded by a Varian Cary 500 Scan UV-vis system. X-ray photoelectron spectroscopy (XPS) measurement was carried out on a RBO upgraded PHI-5000 C ESCA system (Pekin Elmer) using monochromated Al $\text{K}\alpha$ X-rays ($h\nu = 1486.6\text{ eV}$) as a radiation at 250 W. All binding energies were calibrated by using the contaminant carbon ($\text{C}_{1s} = 284.6\text{ eV}$) as a reference.

2.4. Photocatalytic performance tests

Photocatalytic activity tests for the reduction of CO_2 with H_2O were carried out in a stainless steel reactor with the volume of 1500 mL (as shown in Fig. 1). Catalyst powder (0.10 g) was dispersed on the stainless steel omentum which was fixed in the center of reactor. The bottom of glass wool support was moisturized with 5.0 g of deionized water to maintain saturated water vapor in the reactor, and a Xe arc lamp (300 W) with sunlike radiation spectrum was put at the top of quartz window with the distance of 8 cm, which can irradiate the whole catalyst powder. In order to control the reaction temperature, the Xe lamp was placed in a cold trap and the whole reactor system is in the air circulation. Prior to illumination, the reactor was first purged with the $\text{CO}_2 + \text{H}_2\text{O}$ mixture at 100 mL/min for 2 h and then reduced to 20 mL/min for another 1 h to establish an adsorption-desorption balance, and the gas phase volume concentrations of CO_2 and H_2O were controlled at about 95.5% and 4.5%, respectively. After that, the reactor was tightly closed and the Xe lamp was then switched on to start the experiment. The reaction temperature and pressure were maintained at 30°C and 110 kPa, respectively. The gas phase products were taken at various time during the irradiation and analyzed by gas chromatography.

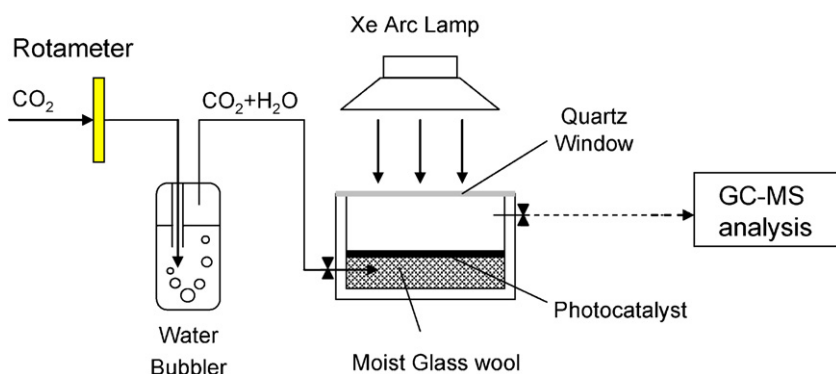


Fig. 1. Schematic of experimental setup for the photocatalytic reduction of CO_2 with H_2O .

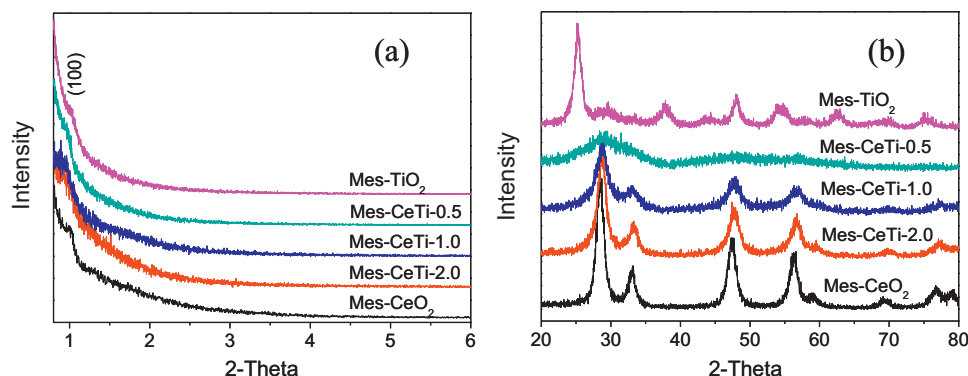


Fig. 2. Low-angle (a) and wide-angle (b) XRD patterns of Mes-TiO₂, Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, and Mes-CeO₂.

3. Results and discussion

3.1. Characteristics of catalysts

The formation of ordered mesoporous CeO₂-TiO₂ composites with different compositions is firstly provided by XRD analysis as shown in Fig. 2. The low-angle XRD patterns in Fig. 2a show that a moderate diffraction corresponding to (100) reflection of the 2D hexagonal (*p6mm*) symmetry can be observed for all samples, indicating the synthesized CeO₂-TiO₂ composites have retained some regular mesostructures although the higher angle two diffractions indexed to (110) and (200) are too weak to be detected, as confirmed by their TEM images (see below). Fig. 2b shows the wide-angle XRD patterns, it can be seen that in addition to an anatase structure observed for pure TiO₂, the structures of pure CeO₂ and other three CeO₂-TiO₂ composites were identified as cubic fluorite phase. It should be noted that the peaks of cubic fluorite phase become broader and the intensity becomes weaker with the decreasing content of CeO₂ in the composites, and sample Mes-CeTi-0.5 with the lowest of Ce/Ti molar ratio exhibits a very broad diffraction peak at $2\theta=24\text{--}37^\circ$ which is ascribed to the overlap of two peaks, indicating amorphous-like structure. The intensity difference and peak broadening of the CeO₂-TiO₂ composites were attributed to a significant reduction of crystallite size. In addition, the peaks corresponding to cubic fluorite phase of composites are shifted toward lower angle with increasing CeO₂ content because of the larger ionic radius of Ce⁴⁺ (0.97 Å) than Ti⁴⁺ (0.61 Å). Energy dispersive X-ray (EDX) analysis showed that the molar ratio of the residual Si in these ordered mesoporous CeO₂-TiO₂ composites was very low (Table 1), indicating most of the silica template has been removed.

The nitrogen adsorption/desorption isotherms and pore size distribution curves of above samples are shown in Fig. 3. All isotherm curves show an obvious uptake of N₂ as a result of capillary condensation in a wide relative pressure (P/P_0) range of 0.55–0.95, which indicates the existence of multiform pore distributions. The pore size distribution obtained from an analysis of desorption branch of the isotherms is shown in Fig. 3b. It can be seen that except Mes-TiO₂ with a relative broad pore size distribution, other four samples possess bimodal pore size distributions centered at about

3.3 and 9.1 nm which are respectively ascribed to the dissolution of the silica walls and the coalescence of unfilled spaces of the template SBA-15 or/and the voids between the small particles. The corresponding textural properties are summarized in Table 1. These ordered mesoporous materials generally have high BET surface areas (134.6–254.1 m²/g) and large pore volumes (0.515–0.832 cm³/g), and both are increased with the decrease of CeO₂ content in composites.

Fig. 4 shows the TEM images of the Mes-TiO₂, Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, and Mes-CeO₂. From the low-magnification TEM images (Fig. 4 left), it can be seen that besides the rodlike morphology, ordered 2D hexagonal mesostructure can be clearly observed for all samples, suggesting thus-prepared materials are well reverse-replica of the template, which is consistent with low-angle XRD results. The high resolution TEM (HRTEM) images in Fig. 4 right show the detail structure of the mesopores, and the pore walls are well constituted by metal oxide nanocrystals with a diameter of several nanometers. The selected area electron diffraction (SAED) patterns are given in the inset of Fig. 4 left. All shows the diffraction rings, indicating that the metal oxide nanocrystals, as building units of the pore walls, are polycrystalline. It should be noted that the diffraction rings for Mes-CeTi-0.5 are not very clear since the semicrystalline structure, which is in agreement with its wide-angle XRD result.

In order to investigate the electronic states of the catalysts, UV-vis diffuse reflectance spectroscopy (DRS) studies were performed on Mes-TiO₂, Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, and Mes-CeO₂ (as shown in Fig. 5). For comparison, a commercial titania powers P25 was also measured and plotted in same Figure. Interestingly, the Mes-TiO₂ displays a slight red shift of light absorption with respect to the commercial P25 which absorbs light with wavelength shorter than 400 nm, this is phenomenon may ascribed to its ordered mesoporous structure, enlarged surface area and multiple scattering enable it to harvest light much more efficiently [40]. The samples of three ordered mesoporous CeO₂-TiO₂ composites show spectral response in a broad visible region ($\lambda=400\text{--}600\text{ nm}$) owing to the photosensitizing effect of CeO₂. It is worthy to note that these CeO₂-TiO₂ composites exhibit a much stronger absorption in both ultraviolet and visible-light regions than that of Mes-TiO₂, this enhanced light-trapping effect is the result of the reflection

Table 1

Textural properties and the energy of band gap (E_g) of Mes-TiO₂, Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, and Mes-CeO₂.

Sample	Ce:Ti molar ratio	Ce:Ti molar ratio (XPS)	Surface area (m ² /g)	Pore size (nm)	Pore volume (cm ³ /g)	Residual Si (mol%)	E_g (eV)
Mes-TiO ₂	0:1	—	254.1	5.4	0.832	1.4	2.81
Mes-CeTi-0.5	1:2	1:2.08	182.4	3.3, 9.1	0.611	1.3	2.16
Mes-CeTi-1.0	1:1	1:0.96	163.1	3.3, 9.1	0.608	1.5	2.20
Mes-CeTi-2.0	2:1	2:1.17	141.9	3.3, 9.1	0.529	1.5	2.24
Mes-CeO ₂	1:0	—	134.6	3.3, 9.1	0.515	1.7	2.64

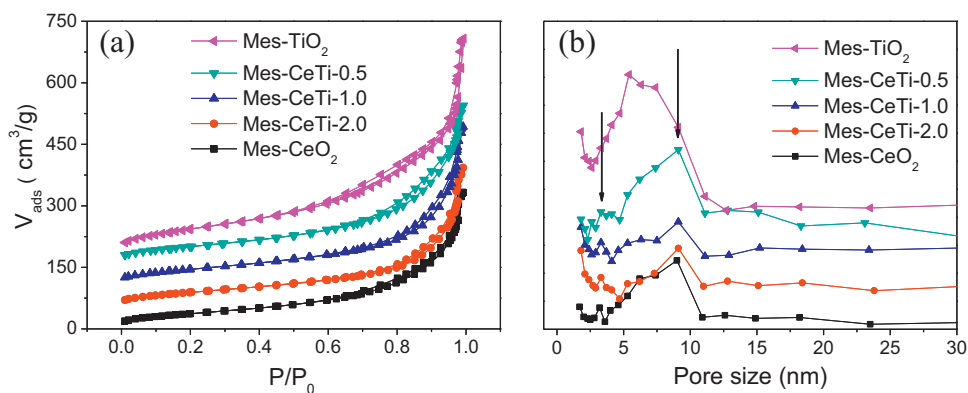


Fig. 3. N₂ adsorption/desorption isotherms (a) and corresponding pore size distribution curves (b) of Mes-TiO₂, Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, and Mes-CeO₂.

or transmission of the light scattered by the ordered mesopores implanted in the body of CeO₂-TiO₂ composites, and a similar effect has been verified in literature [32]. The band energy gap of above samples could be calculated by using $(\alpha h\nu)^n = k(h\nu - E_g)$, where α is the absorption coefficient, k is the parameter that related to the effective masses associated with the valence and conduction bands, n is $1/2$ for a direct transition, $h\nu$ is the absorption energy, and E_g is the band gap energy [41]. Plotting $(\alpha h\nu)^{1/2}$ versus $h\nu$ based on the spectral response in Fig. 5a give the extrapolated intercept corresponding to the E_g value (Fig. 5b). As shown in Table 1, the Mes-TiO₂ exhibits decreased E_g value (2.81 eV) with respect to that of commercial P25 (3.10 eV), which is consistent with the DRS result. Moreover, the optical band gap energies of the Mes-CeTi-0.5, Mes-CeTi-1.0, and Mes-CeTi-2.0 (2.16, 2.20, and 2.25 eV, respectively) display obvious red-shifts compared to that of Mes-TiO₂ (2.83 eV) and Mes-CeO₂ (2.61 eV). The results of this study therefore indicate that the enhanced ability to absorb visible-light of these ordered mesoporous CeO₂-TiO₂ composites makes them a promising photocatalyst for solar-driven applications.

The surface chemical state of Ce, Ti, and O in all samples was also investigated by XPS analysis. Ce 3d spectra of Mes-CeO₂ and three ordered mesoporous CeO₂-TiO₂ composites are shown in Fig. 6a. The labels used in identifying Ce 3d XPS peaks were established by Burroughs et al. [42], where ν and u indicate the spin-orbit coupling 3d_{5/2} and 3d_{3/2}, respectively. The peaks denoted as ν' , ν'' , and ν''' are contributed by CeO₂, which can be respectively assigned to a mixture of Ce IV (3d⁹4f²) O (2p⁴), Ce IV (3d⁹4f¹) O (2p⁵) and Ce IV (3d⁹4f⁰) O (2p⁶). The same peak assignment is applied to u structures. The ν'/u' doublet is ascribed to photoemission from Ce³⁺ cations [43]. Therefore, a mixture of Ce³⁺/Ce⁴⁺ oxidation states exists on the surface of the Mes-CeO₂ and three ordered mesoporous CeO₂-TiO₂ composites, which was in good agreement with the studies of Fang et al. [44] and Gao et al. [45]. The surface concentrations of Ce³⁺ for Mes-CeO₂ and three ordered mesoporous CeO₂-TiO₂ composites evaluated by deconvolution of the XPS spectra are summarized in Table 2, it is worthy to note that Ce³⁺ concentrations in CeO₂-TiO₂ composites are much higher

than that of Mes-CeO₂, suggesting that the introduction of TiO₂ into CeO₂ causes the partial reduction of Ce tetravalent cation [46].

The binding energies of Ti 2p_{1/2} and Ti 2p_{3/2} in both commercial P25 and Mes-TiO₂ are about 464.4 and 458.4 eV, respectively (Fig. 6b). Meanwhile, the presence of CeO₂ in composites shows no significant influence on the Ti2p binding energies which vary at most within ± 0.3 eV. This indicates that Ti exists as Ti⁴⁺ in all samples, which agreed well with the values reported in the literature [44,47]. It can be also observed from Fig. 6b that the Ti 2p XPS peaks became broader after the introduction of CeO₂, which was attributed to the heterogeneous environments of Ti⁴⁺ in CeO₂-TiO₂ composites [44]. The surface Ce/Ti molar ratios derived from XPS for three ordered mesoporous CeO₂-TiO₂ composites are listed in Table 1, it can be seen that the results are in good agreement with the ratios in initial mixture.

Fig. 6c shows the O1s XPS spectra and the deconvolution results of all samples. The O1s XPS spectra appear broad, which can be deconvoluted into three distinct peaks: the one with lower binding energy is assigned to lattice oxygen in the metal oxides, the next is ascribed to chemisorbed oxygen or/and weakly bonded oxygen species, and the third peak at the highest binding energy is due to surface oxygen by hydroxyl species and/or adsorbed water species on the surface [38,45,48]. Based on the area integral of the three peaks respectively corresponding to lattice oxygen (OL), chemisorbed oxygen (OC) and surface oxygen (OS) of O 1s photoemissions, the ratio of OC to OT (OT = OL + OC + OS) for all samples was calculated as shown in Table 2. It can be noticed that the introduction of CeO₂ species can effectively enhance the chemisorbed oxygen on the surface of the CeO₂-TiO₂ composite. Because of the ordered mesostructure and high specific surface area, the Mes-TiO₂ possesses higher ratio of OC/OT than that of commercial P25.

3.2. Photoreduction of CO₂ under simulated solar irradiation

The photocatalytic activity of Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, Mes-TiO₂, Mes-CeO₂, and commercial P25 was evaluated in the reduction of CO₂ with H₂O under simulated solar irradiation. Fig. 7 shows the evolution of main products (CO and CH₄) as functions of irradiation time over all photocatalysts. It can be seen that the yields of CO and CH₄ are increased with the irradiation time for all photocatalysts, and three ordered mesoporous CeO₂-TiO₂ composites exhibit higher photocatalytic activity than Mes-TiO₂ as well as Mes-CeO₂, suggesting that CeO₂ addition can enhance the photocatalytic efficiency of pure mesoporous TiO₂. Moreover, the ordered mesoporous CeO₂-TiO₂ composites show a much higher level of activity than that of commercial P25.

Table 2

Summary of the XPS data for Mes-TiO₂, Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, Mes-CeO₂ and commercial P25.

Samples	Ce ³⁺ /(Ce ³⁺ + Ce ⁴⁺) (%)	OOH/OT (%)
Mes-TiO ₂	–	34.6
Mes-CeTi-0.5	31.1	46.7
Mes-CeTi-1.0	28.9	42.1
Mes-CeTi-2.0	28.8	44.9
Mes-CeO ₂	21.2	40.3
Commercial P25	–	29.1

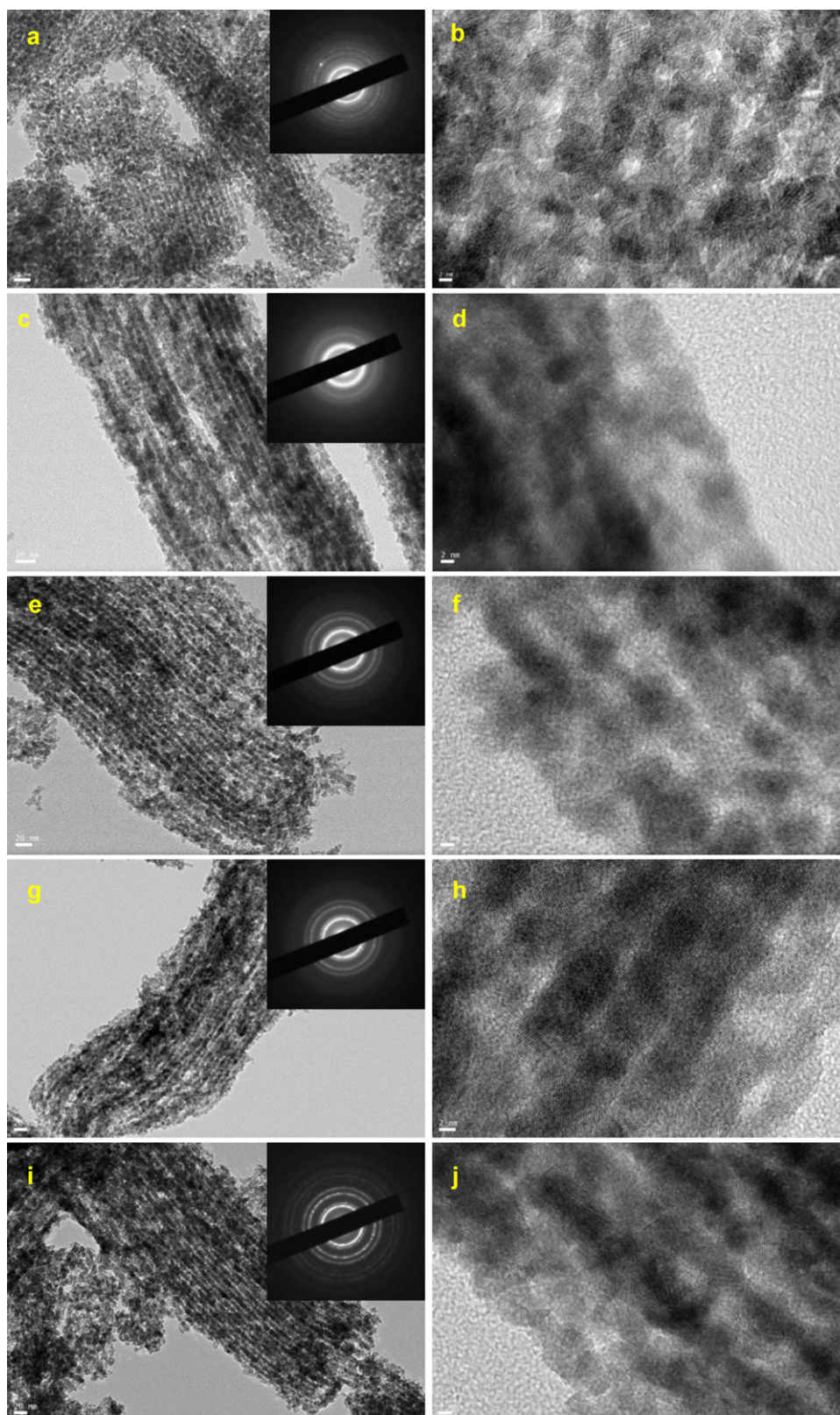


Fig. 4. TEM (left) and HRTEM (right) images of (a, b) Mes-TiO₂, (c, d) Mes-CeTi-0.5, (e, f) Mes-CeTi-1.0, (g, h) Mes-CeTi-2.0, (i, j) Mes-CeO₂.

The enhanced photocatalytic performance for these CeO₂-TiO₂ composites can be ascribed to their unique structure with a variety of favorable properties. First, ordered mesoporous architecture with large surface area and 2D open-pore system makes facile the

reactant diffusion into the bulk of catalyst and hence provide fast intraparticle molecular transfer. Second, the introduction of CeO₂ species in composites can effectively extend the spectral response from UV to visible area owing to the CeO₂-photosensitization. In

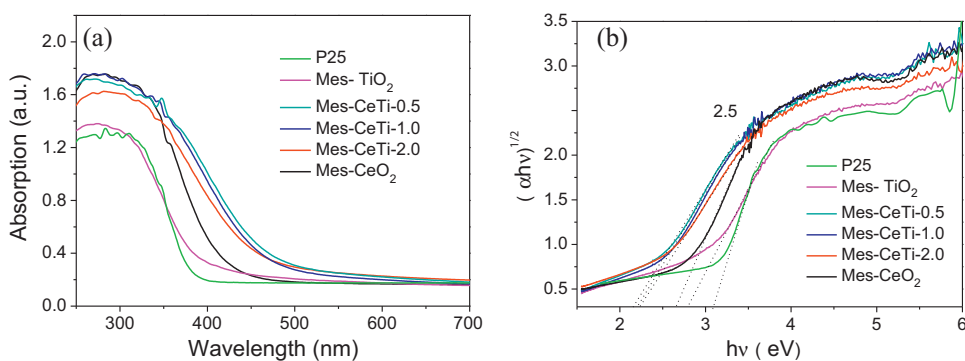


Fig. 5. (a) UV-vis diffuse reflectance spectra (DRS) and (b) the optical absorption edges of the Mes-TiO₂, Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, Mes-CeO₂ and commercial P25.

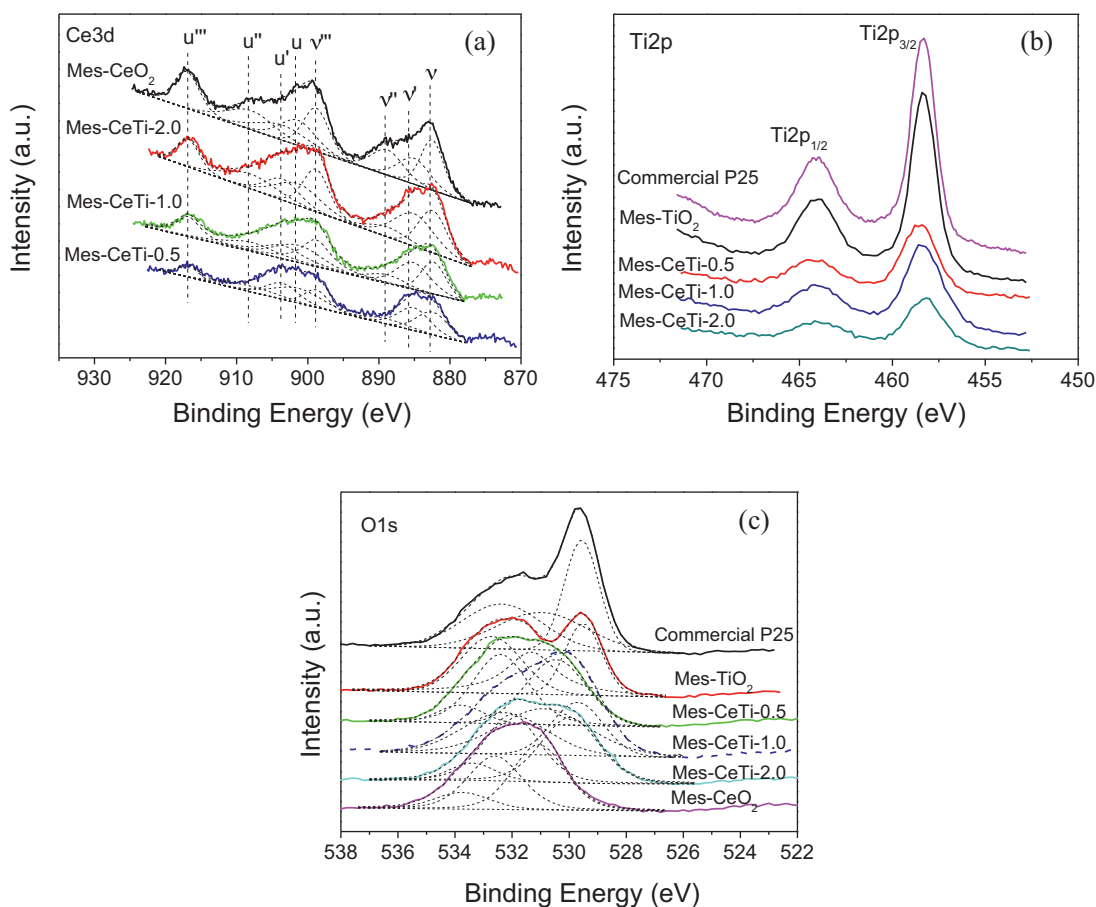


Fig. 6. (a) Ce 3d, (b) Ti 2p, and (c) O 1s XPS spectra of the Mes-TiO₂, Mes-CeTi-0.5, Mes-CeTi-1.0, Mes-CeTi-2.0, Mes-CeO₂ and commercial P25.

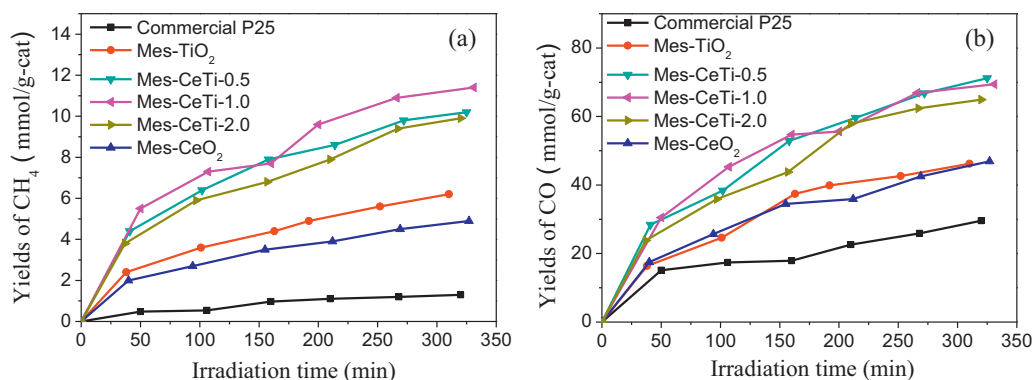


Fig. 7. Yields of CH₄ (a) and CO (b) as functions of irradiation time over all photocatalysts.

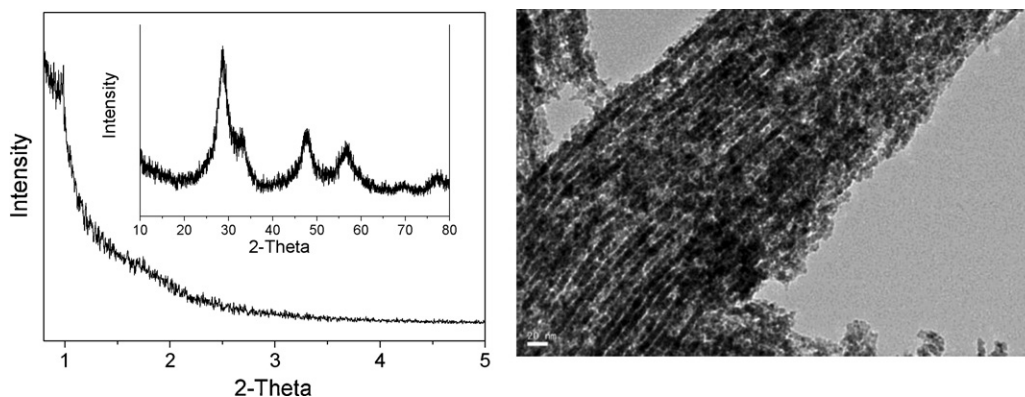


Fig. 8. XRD patterns (left) and TEM image (right) of the representative Mes-CeTi-1.0 sample after photocatalytic test.

our experiment, the 300 W Xe arc lamp excitation spectrum contains about 50% visible-light which is greatly contributed to the high photocatalytic efficiency for the ordered mesoporous CeO_2 - TiO_2 composites in the reduction of CO_2 with H_2O . At the same time, the photogenerated electrons in the TiO_2 can drift to the CeO_2 under the inner electric field between CeO_2 and TiO_2 because of the energy band bending in space charge region, it is more helpful for the separation of photogenerated electron-hole pairs in TiO_2 , resulting in the improvement of photocatalytic activity under solar irradiation. Finally, XPS analysis confirms that the CeO_2 addition can greatly enhance the surface chemisorbed oxygen species of the ordered mesoporous CeO_2 - TiO_2 composites, these oxygen species can easily capture electrons and yield surface oxygen radicals with excellent reduction capability. Also, the existence of mixture of $\text{Ce}^{3+}/\text{Ce}^{4+}$ oxidation states on the surface of CeO_2 - TiO_2 composites denote that the partial metal in composites is not fully oxidized, so that Ce^{3+} can interact with holes and prevent the combination of photogenerated electrons and holes, resulting in a higher quantum efficiency of photocatalytic reaction.

In addition, in order to investigate the structural stability of the ordered mesoporous CeO_2 - TiO_2 composites in photocatalytic reaction in the presence of H_2O , we have characterized the structure of these ordered mesoporous CeO_2 - TiO_2 composites after photocatalytic test by XRD and TEM analysis. It was found that the structure of these used CeO_2 - TiO_2 composites hasn't changed when compared with that of the corresponding fresh ones, indicating a good structural stability. Fig. 8 shows the XRD patterns and TEM image of the representative Mes-CeTi-1.0 sample after photocatalytic test.

4. Conclusions

In conclusion, we have demonstrated that ordered mesoporous CeO_2 - TiO_2 composites with 2D hexagonal structure and varied compositions were synthesized via a simple nanocasting route. XRD, nitrogen sorption, and TEM results reveal that the obtained CeO_2 - TiO_2 composites have ordered 2D hexagonal mesostructures with high specific surface area and hierarchical porosity. Introduction of CeO_2 species to TiO_2 can effectively extend the spectral response from UV to visible area and enhance the surface chemisorbed oxygen species of the ordered mesoporous CeO_2 - TiO_2 composites, which are confirmed by UV-vis and XPS analysis. Due to the special composition and structure, the ordered mesoporous CeO_2 - TiO_2 composites exhibited an excellent photocatalytic activity in the photoreduction of CO_2 with H_2O under simulated solar irradiation.

Acknowledgments

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